# Structure and Stability of Liquid Aluminium-Zinc Alloys

R. KUMAR, C. S. SIVARAMAKRISHNAN National Metallurgical Laboratory, Jamshedpur-7, India

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The structure of liquid aluminium-zinc alloys was investigated by the Kumar-Samarin centrifuging technique, and it was found that liquid alloys falling outside the composition range in which there is solidus inflection are stable and consist of a colloidal dispersion of clusters, while those within that range are incipiently immiscible.

# 1. Introduction

Investigating the structure of liquid eutectictype alloy systems with the help of the Kumar-Samarin technique of centrifuging liquid metals, Kumar and Sivaramakrishnan [1, 2] showed that the liquid solution, over the entire range of composition in binary eutectic systems, may not be homogeneous and stable in the presence of a characteristic inflection in the form of the liquidus. They showed that in liquid lead-cadmium and lead-antimony alloys, (i) liquids outside the composition range of the liquidus inflection on either end of the binary phase diagram are stable and consist of colloidal dispersions of solute- and solvent-rich clusters; (ii) the essential immiscibility existing in the solid state is extended in rudimentary form in the liquid state in the composition range of the inflection, and a region of incipient immiscibility exists above the liquidus on melting; (iii) the incipiently immiscible liquid can be separated into its two conjugate phases by centrifuging in the liquid state. The region of incipient immiscibility was not found to exist in the simple eutectic system Pb-Sn in which the liquidus was not inflected [3].

To study the structure and stability of the liquid in systems which show more pronounced inflection in the solidus than in the liquidus, it was decided to extend the work to the Al-Zn system in which the solidus exhibits an inflection in the composition range 20 to 60 at. % zinc. Just as an inflection in the liquidus is often considered to foreshadow immiscibility in the liquid state, so a region of two solid solutions 1008

exists at lower temperatures below the region of the inflected solidus. The existence of two solid solutions is deduced from discontinuities in the lattice parameter versus composition curve at elevated temperatures [4].

# 2. Experimental

Details of the experimental techniques and of the statistical analysis of the results have been described earlier [1]. Alloys containing 16, 40, 50 and 90 at. % Zn respectively were investigated in the temperature range 450 to 750° C, always above the liquidus. Only the alloys with 16 and 90 at. % Zn fell outside the region of solidus inflection. Samples of about 60 g each of the alloys were centrifuged in graphite crucibles at two speeds of rotation for various times and at different temperatures. The experimental conditions are summarised in table I. During centrifuging the heavier zinc-rich clusters migrated to the outer end of the crucible, resulting in the setting up of concentration gradients in alloys with 16 and 90 at. % zinc as shown in figs. 1 and 2. But the two alloys with 40 and 55 at. % Zn did not behave in this manner.

### 2.1. Cluster Size

In figs. 1 and 2, best-fitting lines are drawn on the basis of a least square analysis. The statistical correlation coefficient and the probability levels were also calculated in each case and are recorded in table I along with their confidence ranges on the concentration gradient for 80 to 95% probability. These calculations show that in general, the correlation between the logarithm

Comp	osition Centrifugal	Temperature	Concentra-	Correla-	No. of	Probability Confidence range $(\pm)$ as			
at. %	Zn force $\times \mathbf{g}$	°C	tion gradient	tion coefficient	observa- tions	level %	concenti 80%	ration gra 90%	adient at 95%
16.0	40	650	0.0075	0.8658	9	>95			0.0038
		750	0.0164	1.0000	9	100	_	<u> </u>	*
	70	650	0.0080	0.6530	9	>80	0.0056		
		750	0.0145	0.7616	9	>95			0.0118
90.0	40	450	0.0189	0.7048	9	>90	_	0.0152	~
		550	0.0478	0.5349	.9	>60	† <b>0</b> .0475	_	
	70	450	0.0232	0.8038	8	>95			0.0073
		550	0.0583	0.4592	9	>70	<b>‡0.0470</b>		

TABLE I Statistics of experimental data of AI-Zn systems

\*100% confidence range; †value at 60% confidence range; ‡value at 70% confidence range



Figure 1 Chemical analysis of Al-16 at. % Zn alloys.



Figure 2 Chemical analysis of AI-90 at. % Zn alloys.

of concentration and the distance, is justified by rigorous statistical analysis, but the correlation is poor when centrifuging is done at high super-heat, as in the case with the eutectic aluminium-zinc alloys (90 at. % Zn) at 550° C. The high volatility of zinc contributes to the 1009

Temperature	Al	Zn	Al-16Z	n Al-90Zn
450	2.37*	6.66		2.827
550	2.37	6.55		2.808
650	2.37	6.50	2.965	
750	2.31	6.40	2.894	

TABLE II Densities of AI, Zn and their liquid alloys

\*Indicates density in the solid state.

relatively poor correlation; better correlation would, however, result if observations of individual runs are considered singly as shown by dashed lines in fig. 2.

The cluster radii were calculated, as was done earlier [1], by the sedimentation equilibrium equation using published values [5] of the density of aluminium and zinc, as summarised in table II. The calculated cluster size in the two aluminium-zinc liquid alloys with 16 and 90 at. % Zn respectively is recorded in table III. The extremes of cluster size within the confidence range of the concentration gradient are also indicated by super- and subscripts in the table. It is noted that the cluster size depends on composition and temperature.

#### 2.2. Region of Incipient Immiscibility

Alloys with 40 and 50 at. % Zn decomposed into two conjugate phases on centrifuging between 550 and 750° C. The results of the chemical analysis in the two phases are given in figs. 3 and 4. It is noted that instead of a gradual concentration gradient being obtained in the solidified ingot, the liquid divided itself into two predominant liquids having their compositions near the two limits of the inflection.

#### 3. Conclusion

The investigation shows that (i) liquid aluminiumzinc alloys are stable in the composition ranges 0 to 20 and 60 to 100 at. % Zn, and consist of



Figure 3 Chemical analysis of AI-40 at. % Zn alloys.



*Figure 4* Chemical analysis of AI-50 at. % Zn alloys. 1010

Com- position at.% Zn	Centri- fugal force $\times g$	Tempera- ture °C	Calculated values of cluster, Å, size at mean, minimum and maximum slopes Al Zn			
	40	650	28	22 32	16	12 18
16	40	750	38		21	
	70	650	24	16 29	13	9 16
		750	30	17 37	17	10 20
	40	450	40	22 47	19	11 23
00		550	56	10 70	27	5 34
90	70	450	34	30 38	17	15 19
		550	50	29 60	24	14 30

TABLE III Cluster size in AI-Zn systems

colloidal dispersion of aluminium- and zinc-rich clusters and, (ii) the liquid in the composition range of the inflection in the solidus consists of an incipiently immiscible solution of two conjugate liquid phases which can be separated by centrifuging.

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